

Electrochemical control of thermal conductivity in thin films

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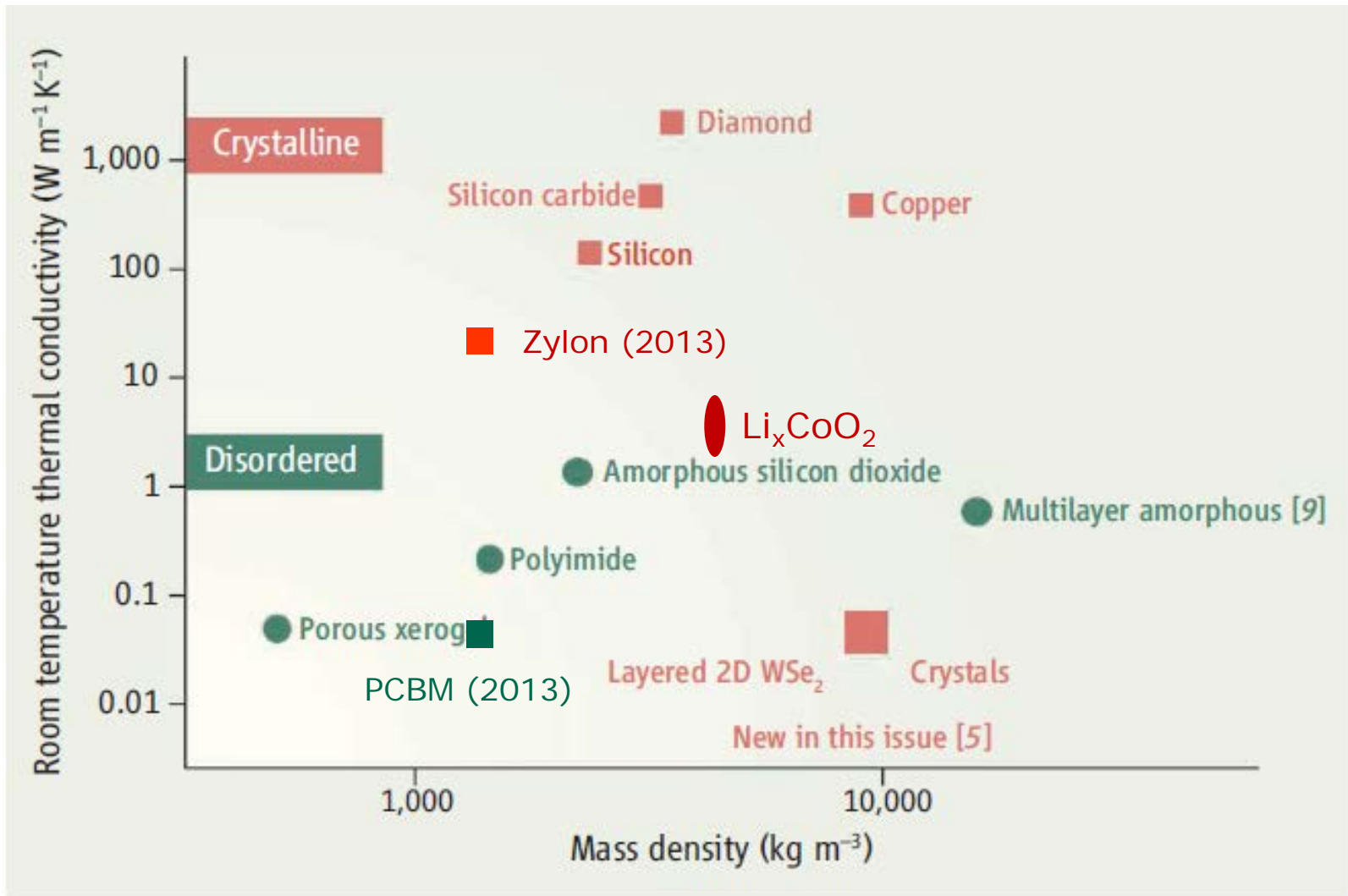
*International Institute for Carbon Neutral Energy
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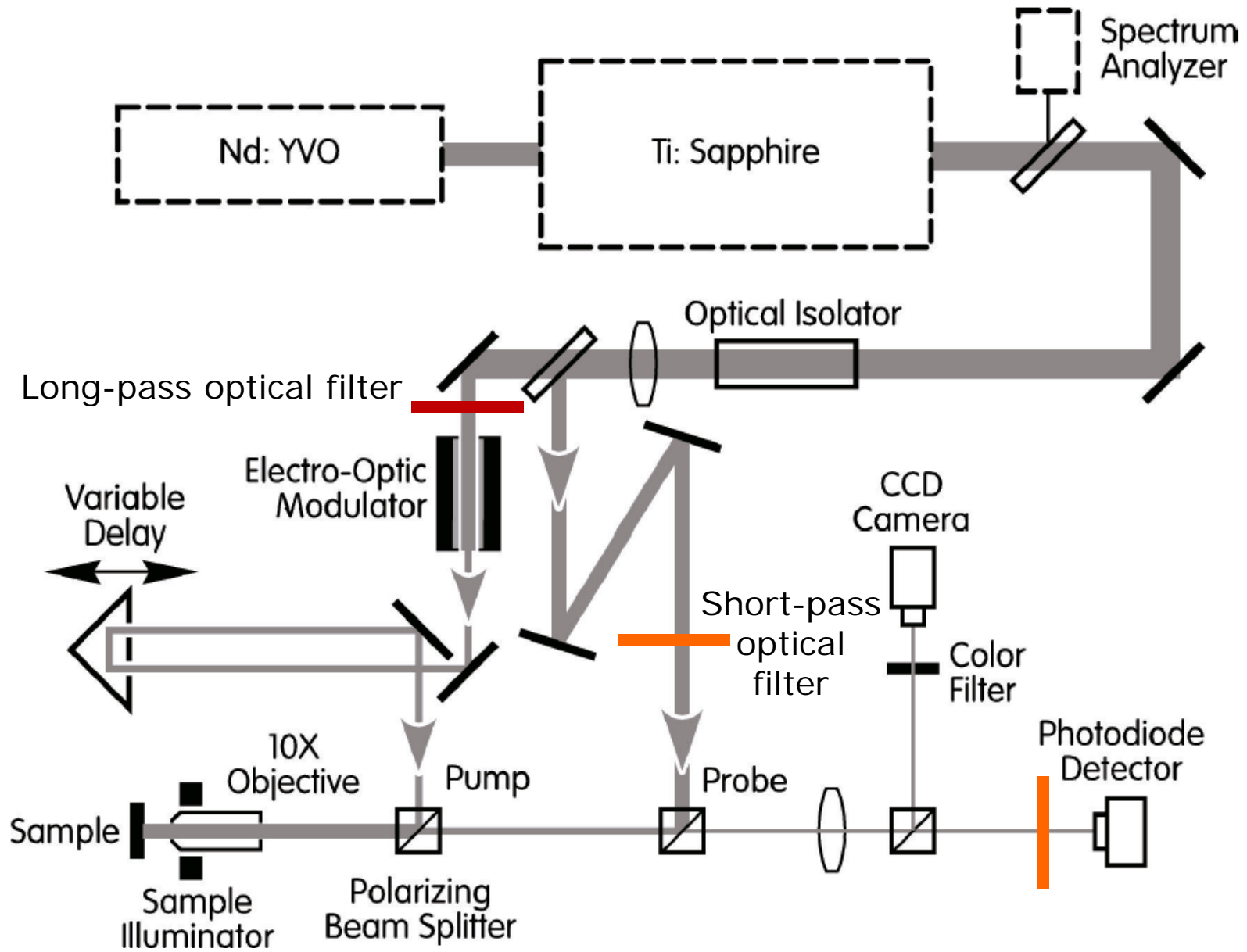
- Thermal conductivity and measurement by time-domain thermoreflectance (TDTR)
- Big picture goals of our work:
 - Understand and push the limits of thermal conductivity in various classes of materials
 - enhance thermal function in materials, e.g., abrupt changes in conductivity, actively controlled conduction, more efficient heat pumping.
- Electrochemical modulation of the thermal conductivity of Li_xCoO_2
 - Materials science and phenomenology
 - Materials physics

Thermal conductivities of dense solids span a range of 40,000 at room temperature

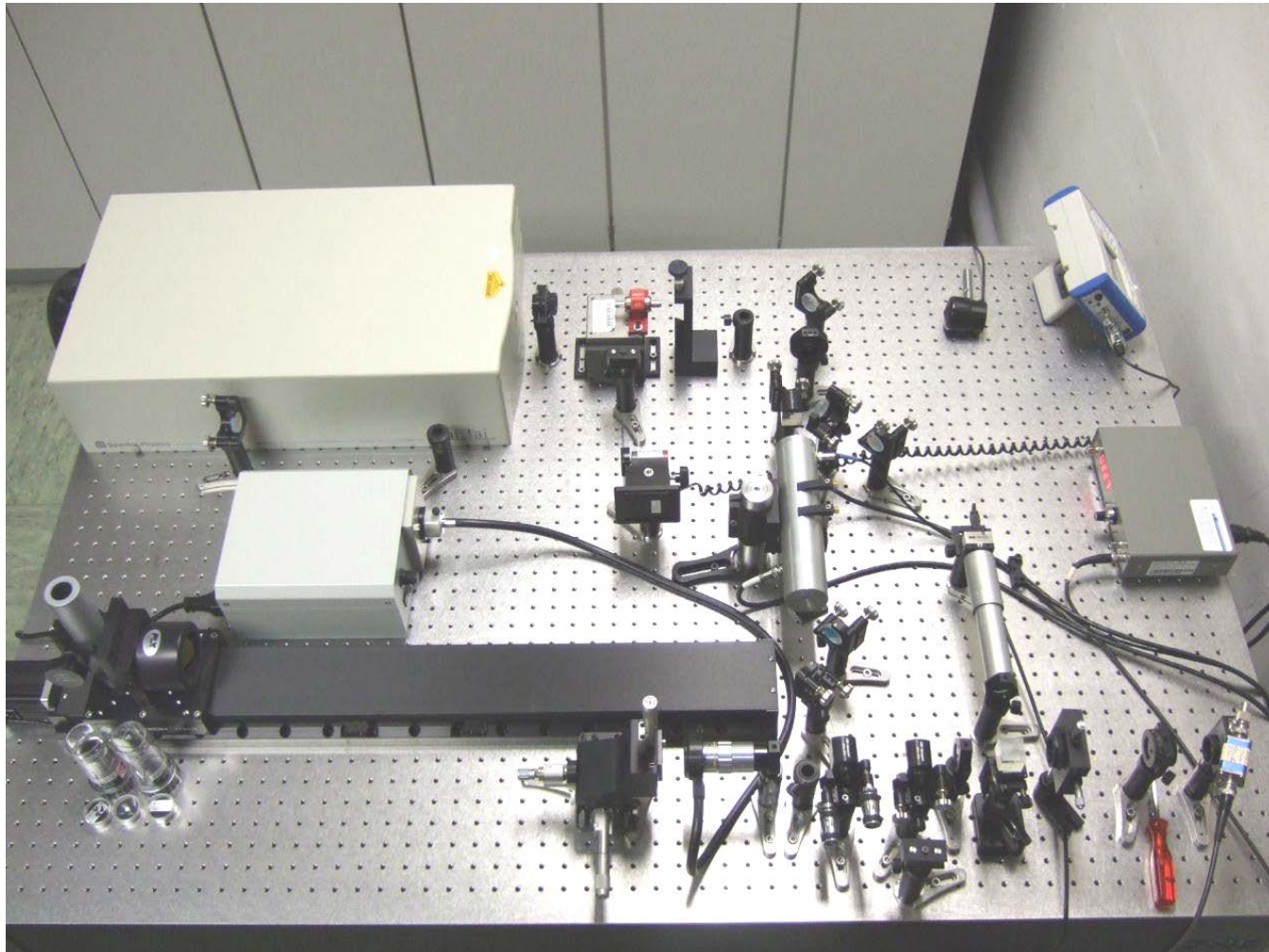


Adapted from Goodson, *Science* (2007)

Time-domain thermoreflectance



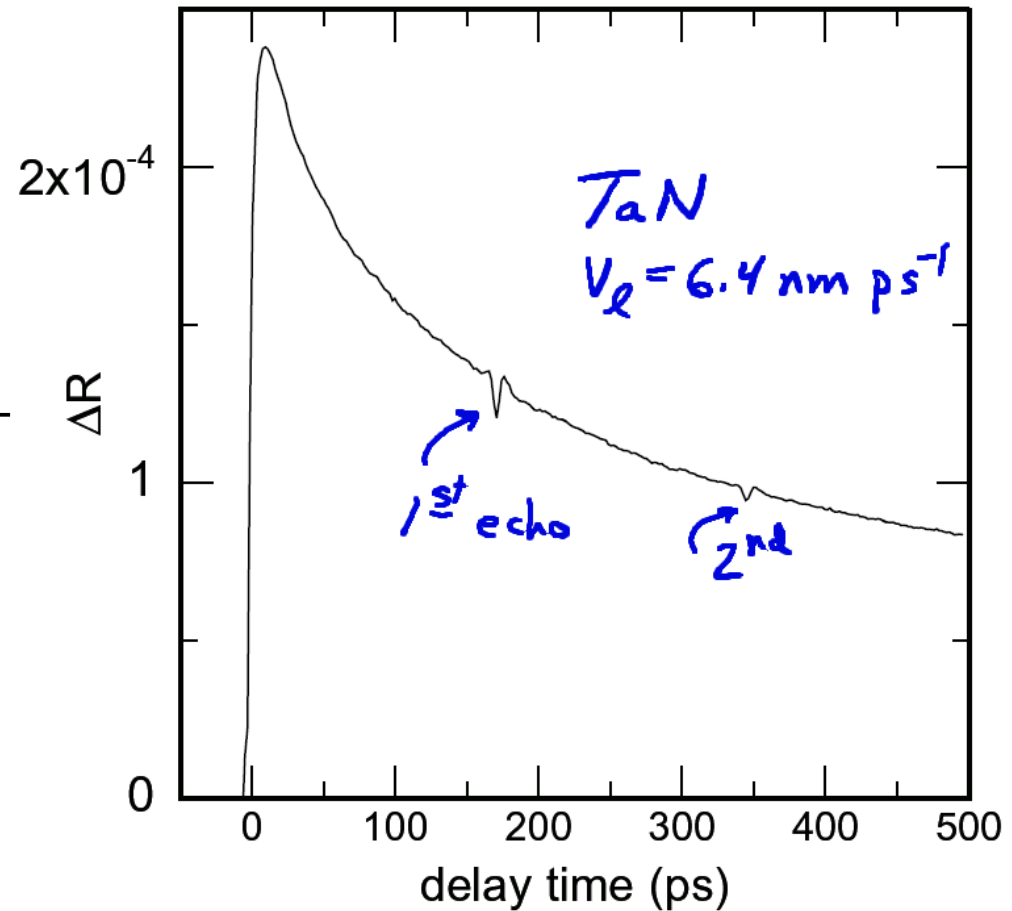
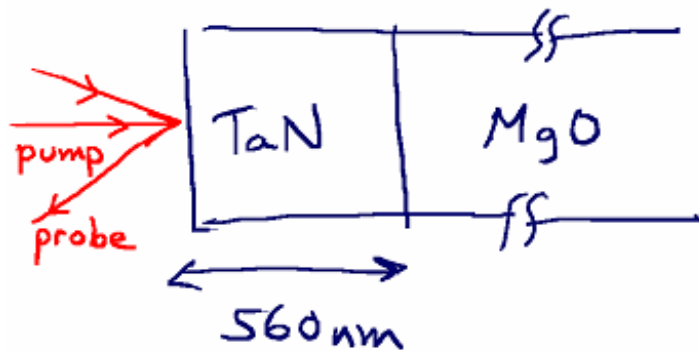
Time-domain thermoreflectance



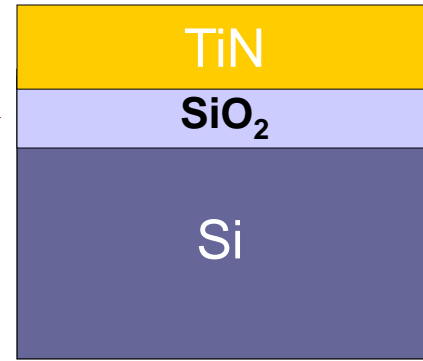
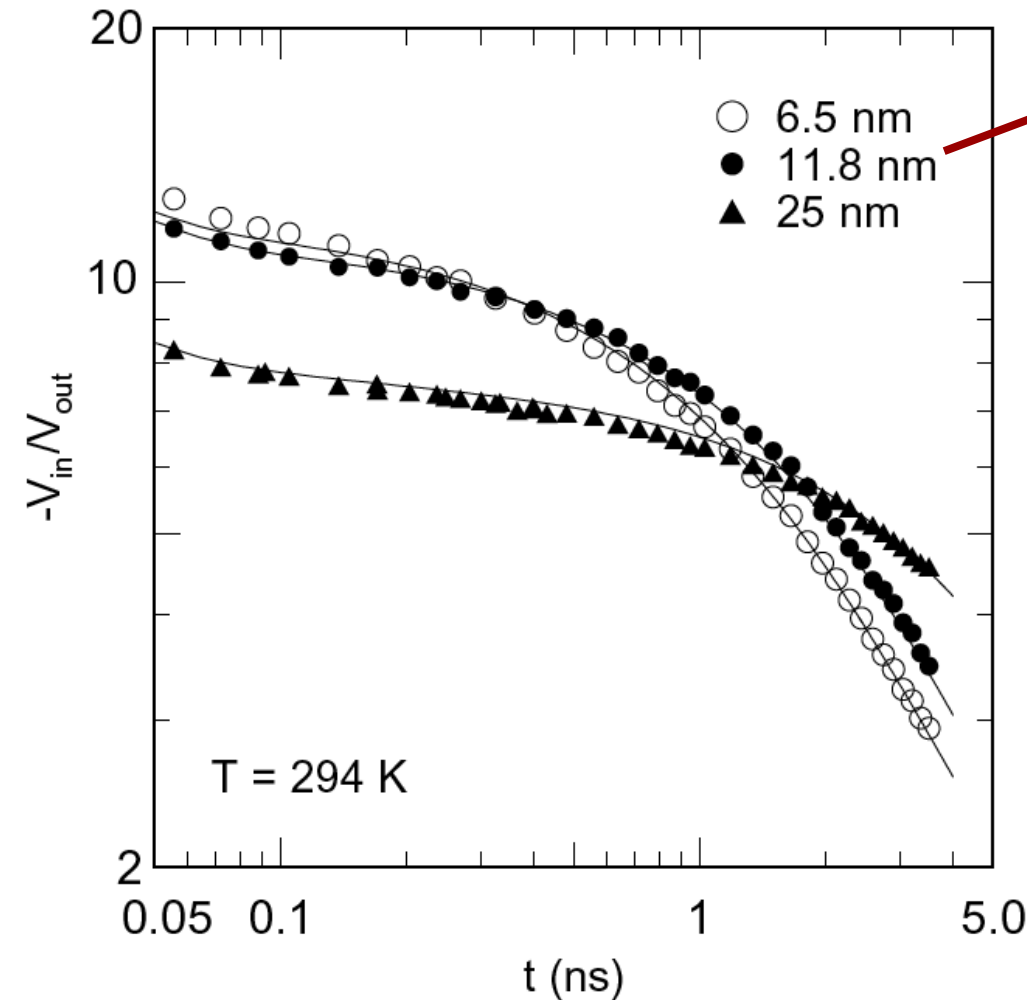
Clone built at Fraunhofer Institute for Physical Measurement, Jan. 7-8 2008

psec acoustics and time-domain thermoreflectance

- Optical constants and reflectivity depend on strain and temperature
- Strain echoes give acoustic properties or film thickness
- Thermoreflectance dR/dT gives thermal properties

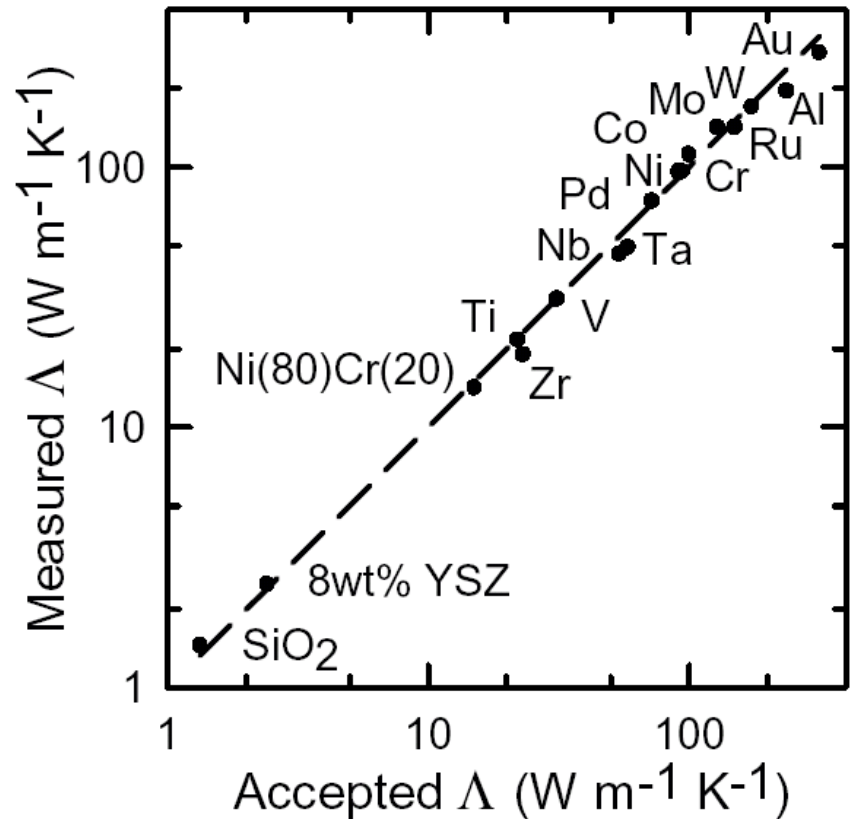
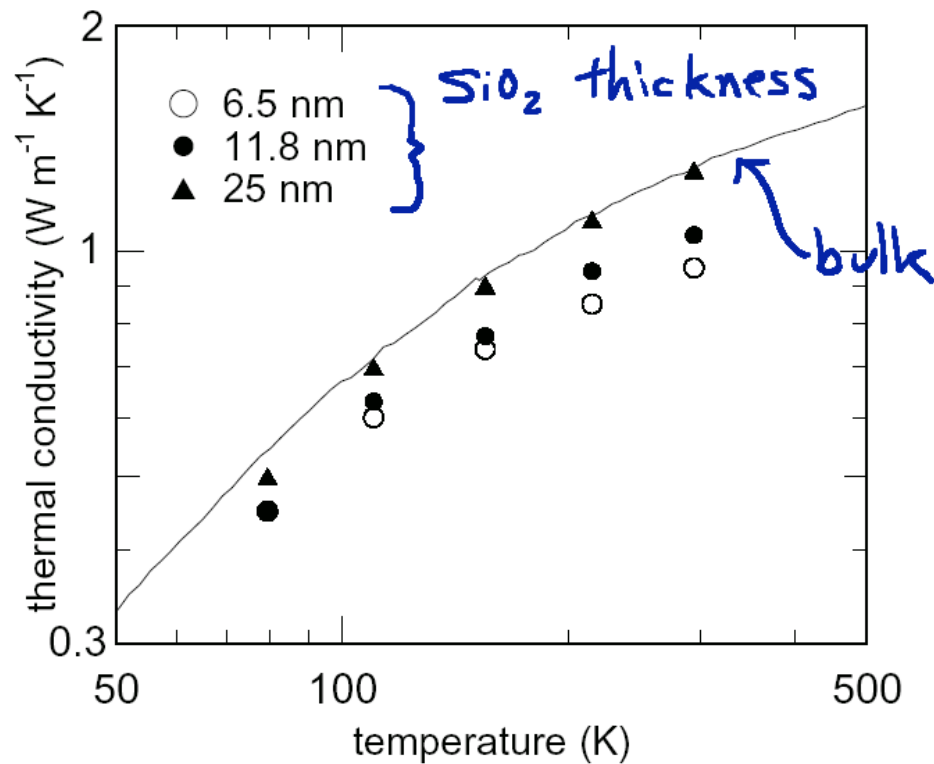


Time-domain Thermoreflectance (TDTR) data for TiN/SiO₂/Si

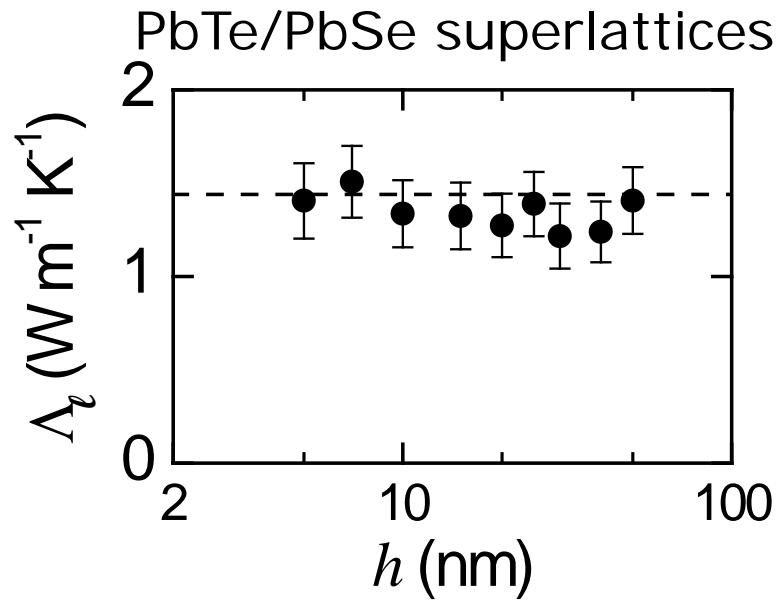


- reflectivity of a metal depends on temperature
- one free parameter: the "effective" thermal conductivity of the thermally grown SiO₂ layer (interfaces not modeled separately)

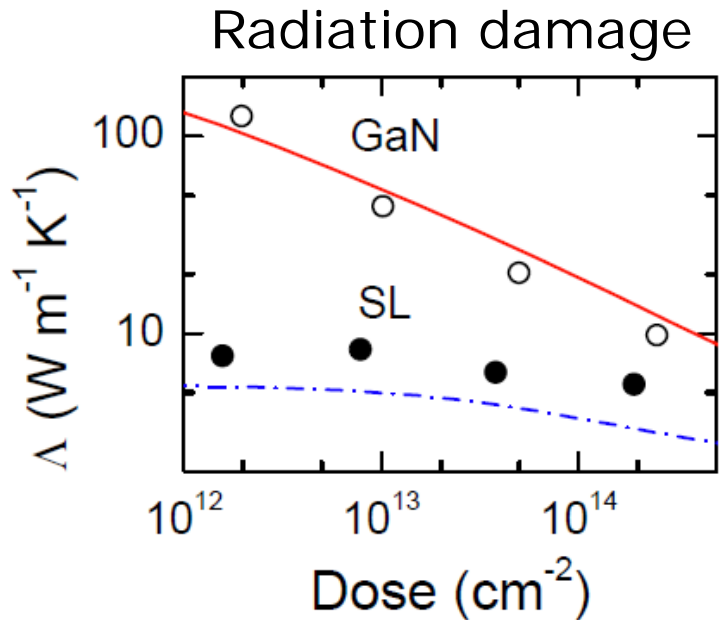
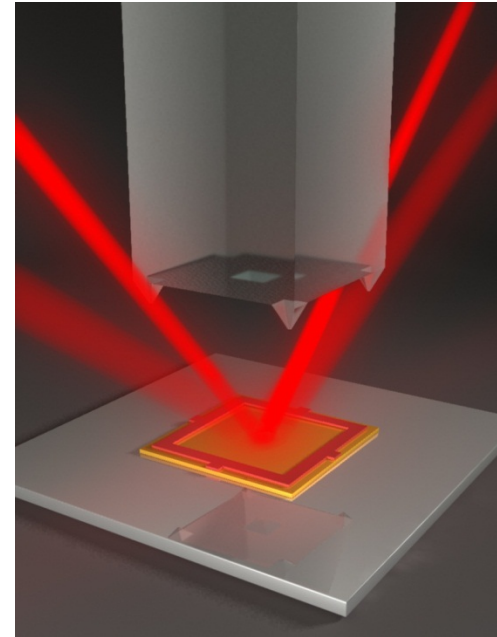
TDTR: validation experiments



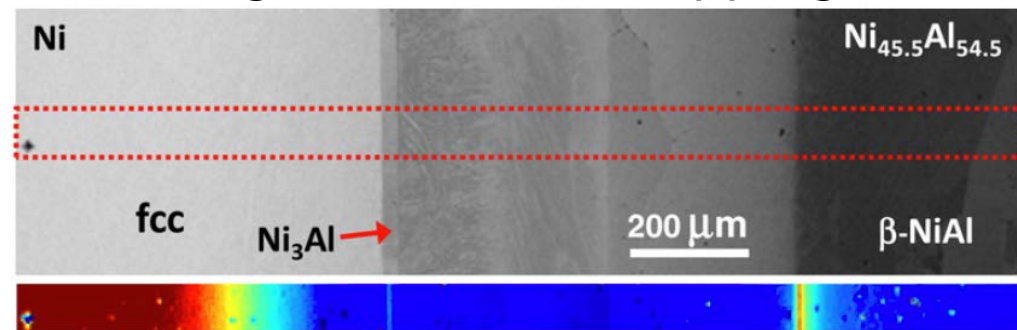
TDTR: Flexible, convenient, and accurate



Transfer-printed interfaces

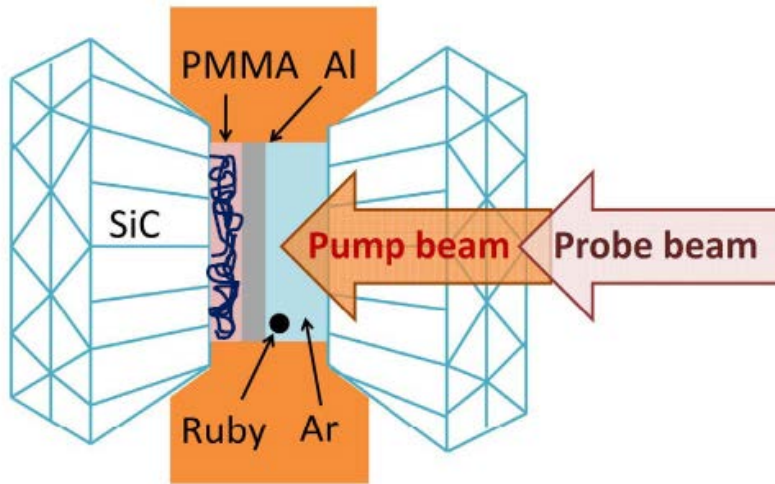


High resolution mapping

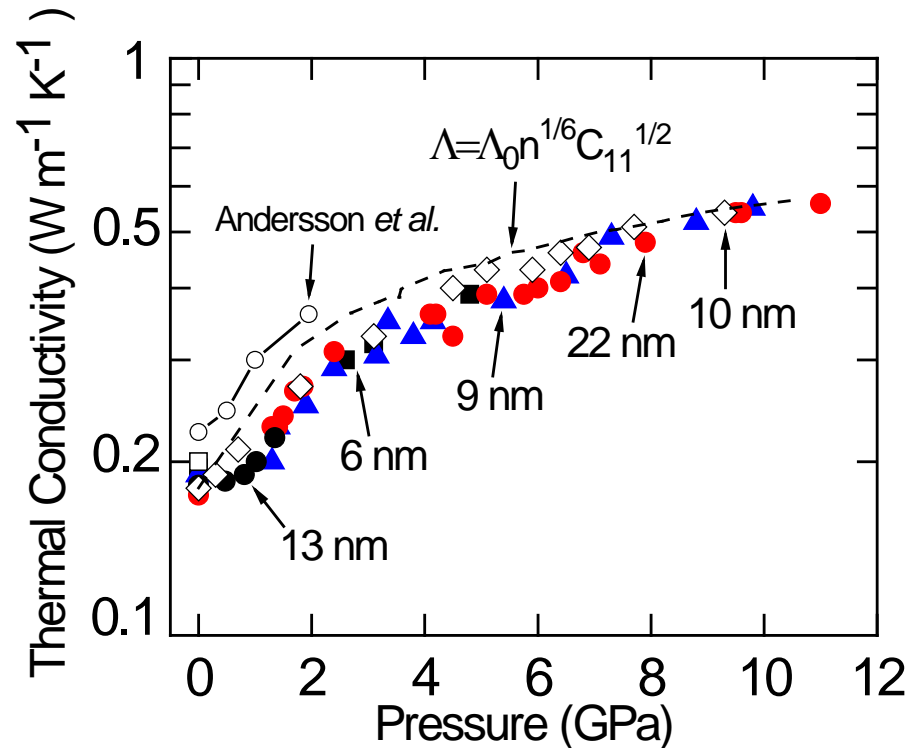


TDTR is all optical method: adaptable to “extreme” environments such as high pressure

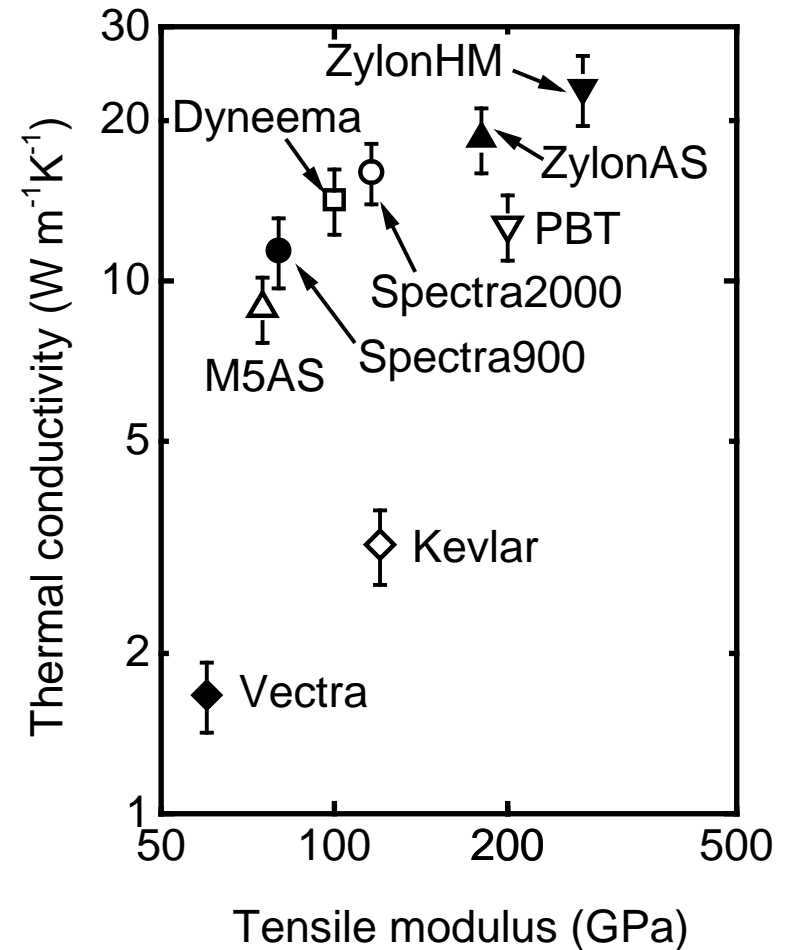
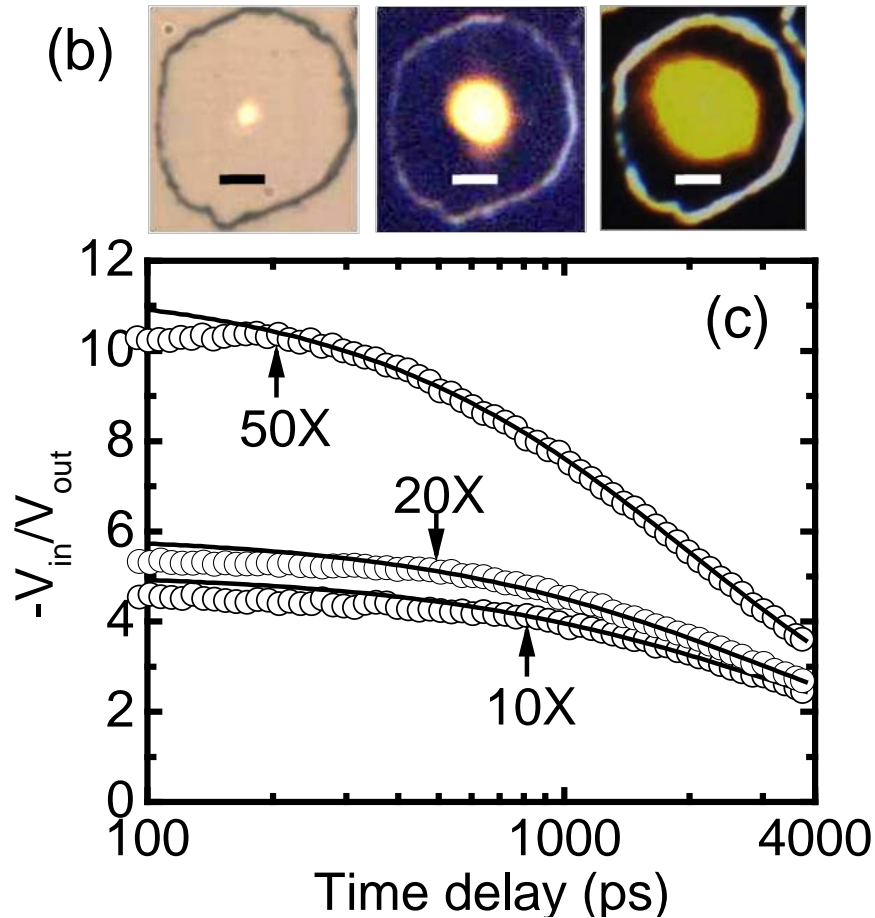
Diamond anvil cell



Thermal conductivity of PMMA is independent of thickness and agrees well with the predicted scaling with $(C_{11})^{1/2}$

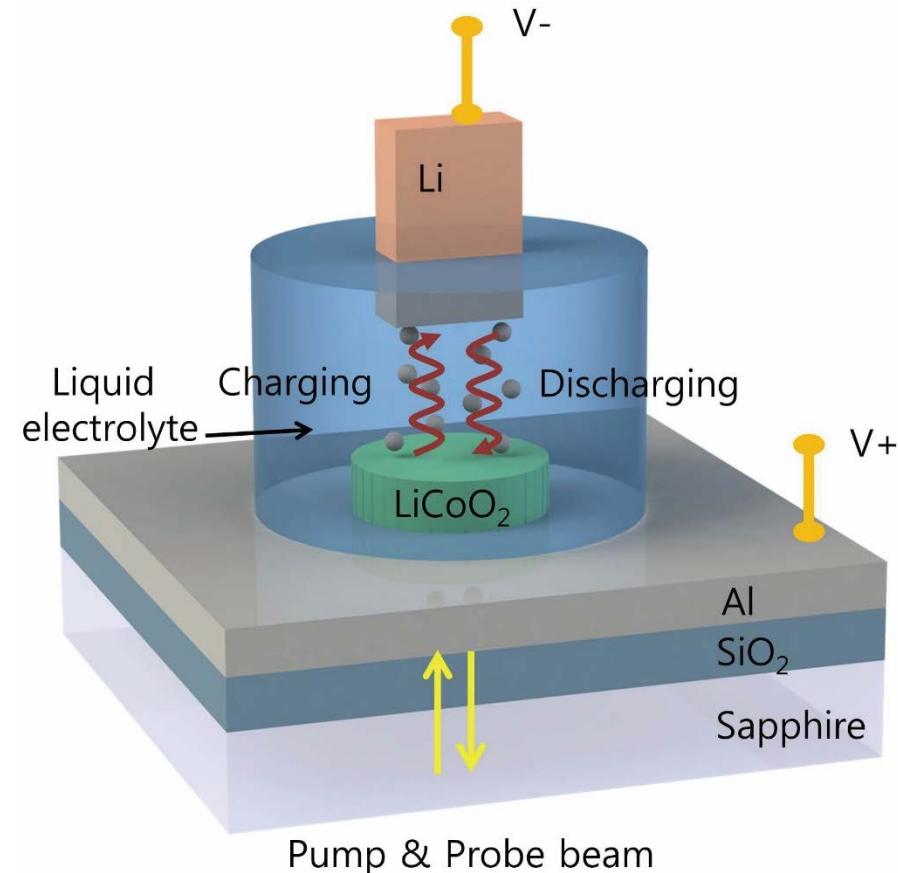


High throughput measurements of polymer fibers by time-domain thermoreflectance



Electrochemical modulation of thermal conductivity of Li_xCoO_2

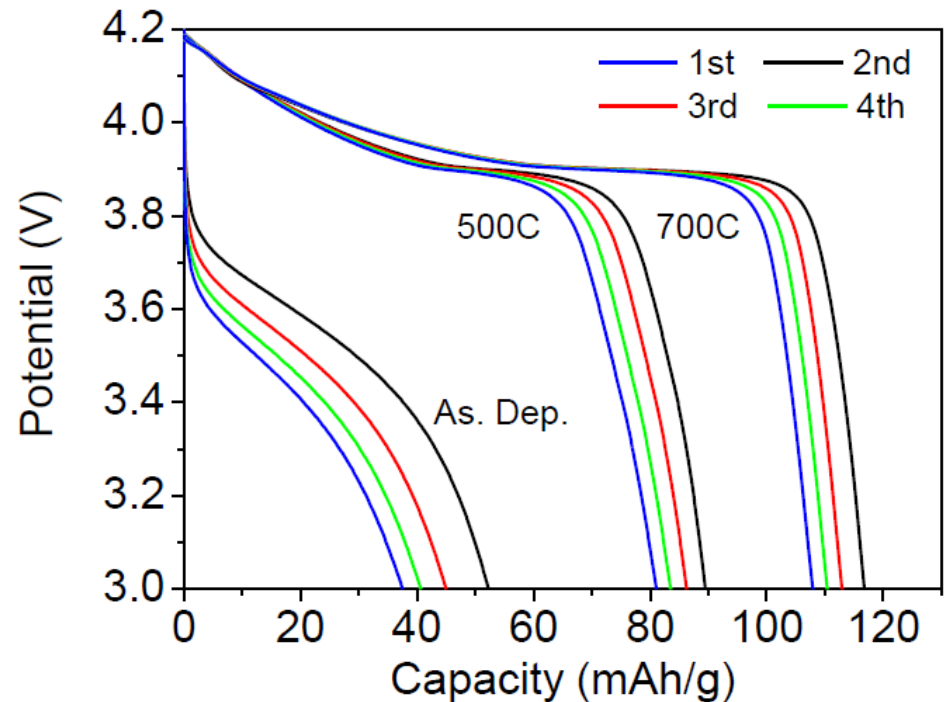
- Polycrystalline thin film prepared by sputter deposition and annealing
- Real-time measurement by TDTR and picosecond acoustics.
 - ✓ Thermal conductivity $3.6 \rightarrow 5.4 \text{ W m}^{-1} \text{ K}^{-1}$
 - ✓ Elastic modulus $220 \rightarrow 300 \text{ GPa}$
 - ✓ Ex-situ thermal conductivity contrast as large as a factor of 2.7



Sputter deposit Li_xCoO_2 and anneal in air

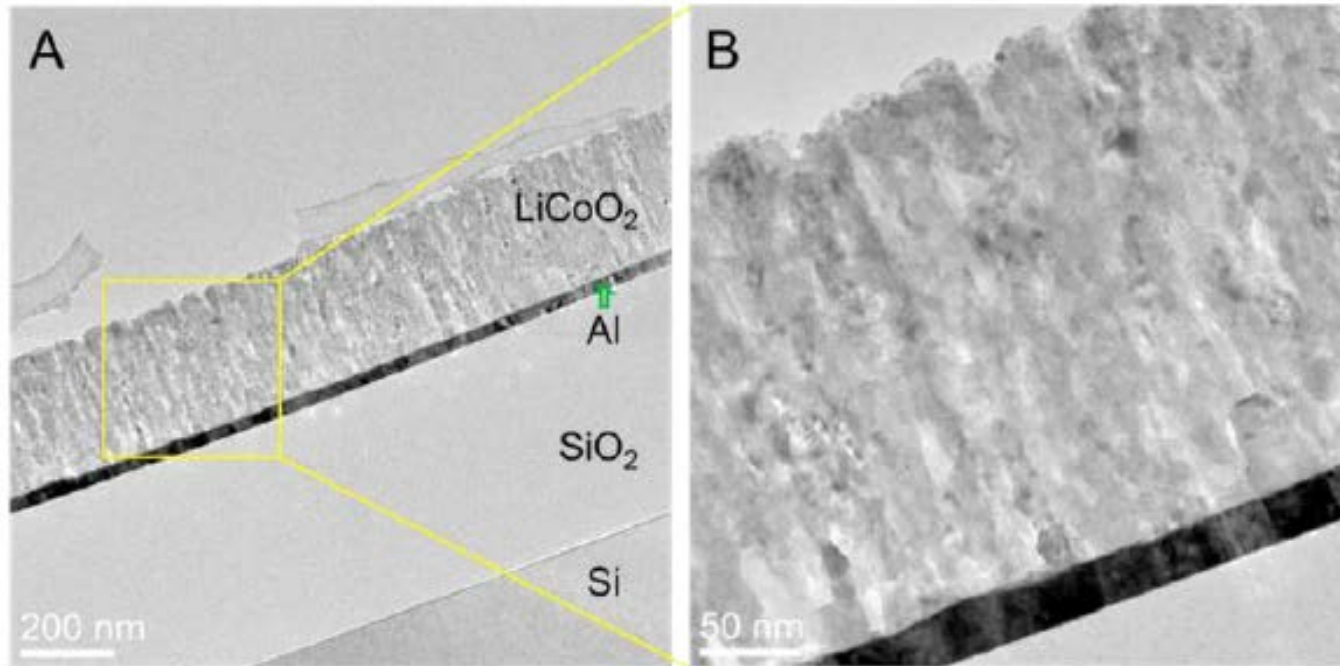
- TDTR works best with Al transducer.
 - Limit annealing temperature of samples for in-situ studies to 500°C

500 nm Li_xCoO_2 ; 0.3C rate



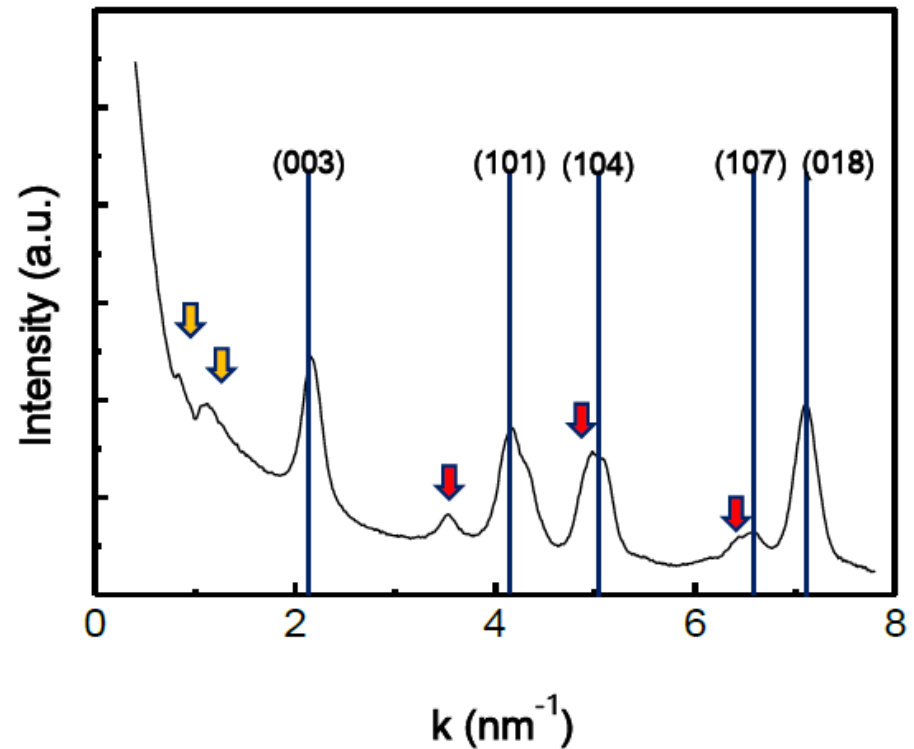
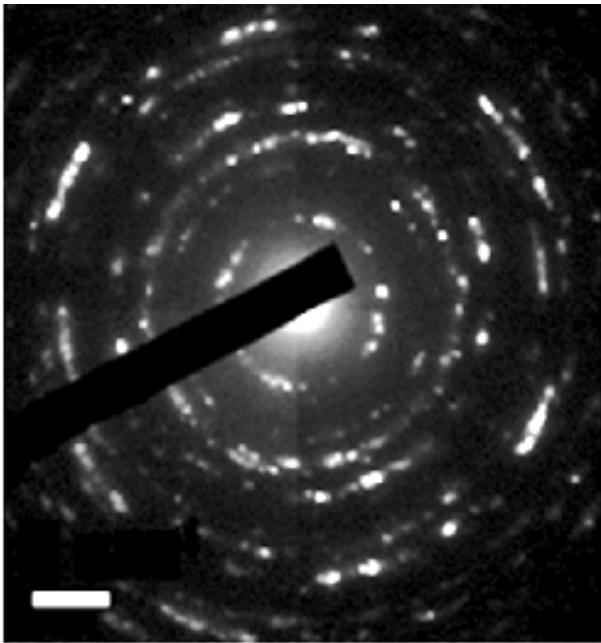
Characterize microstructure by electron microscopy

- After annealing at 500C in air
- Nanocrystalline, dense microstructure



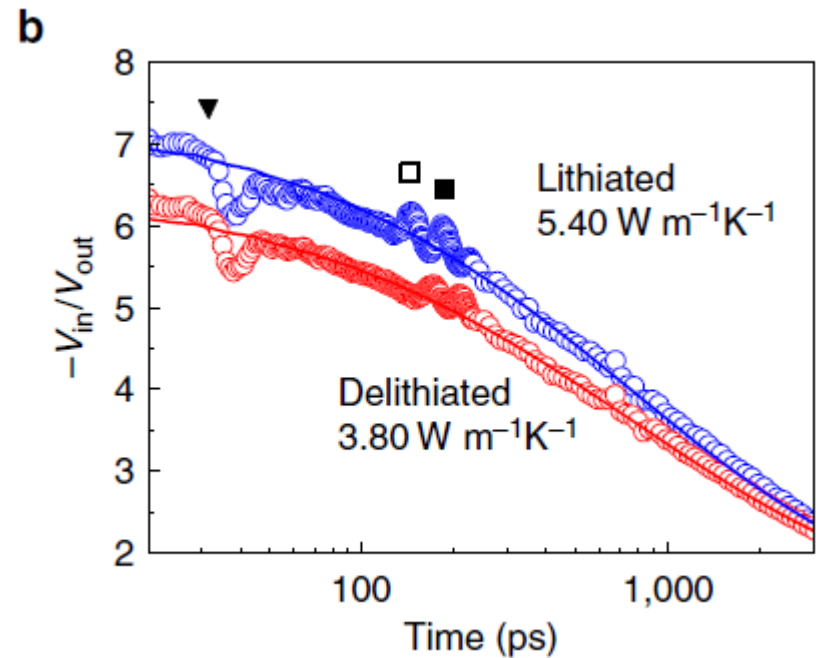
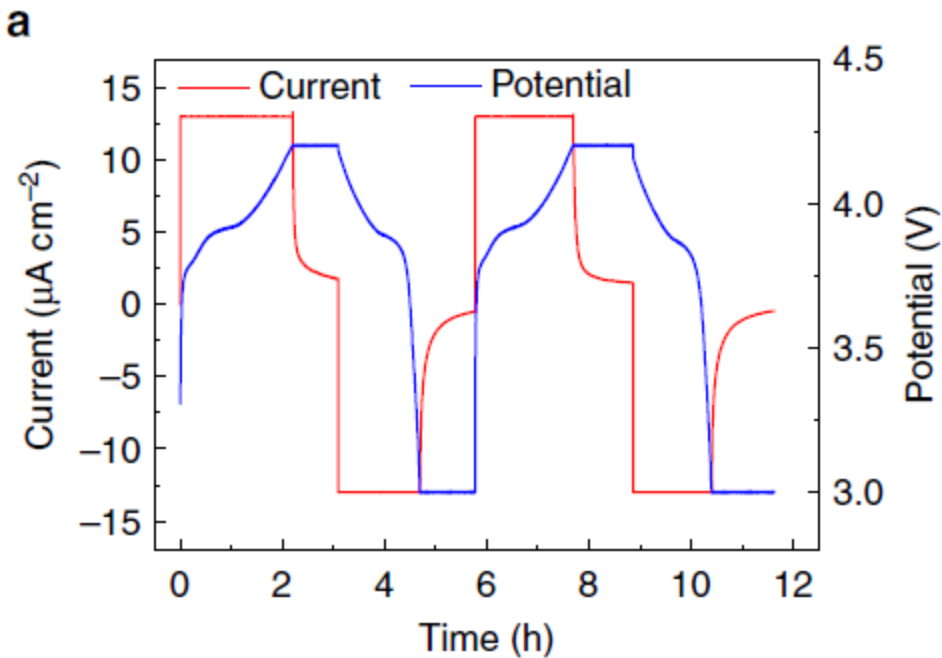
Characterize microstructure by electron diffraction

- No strong texture; would eventually like to study textured films



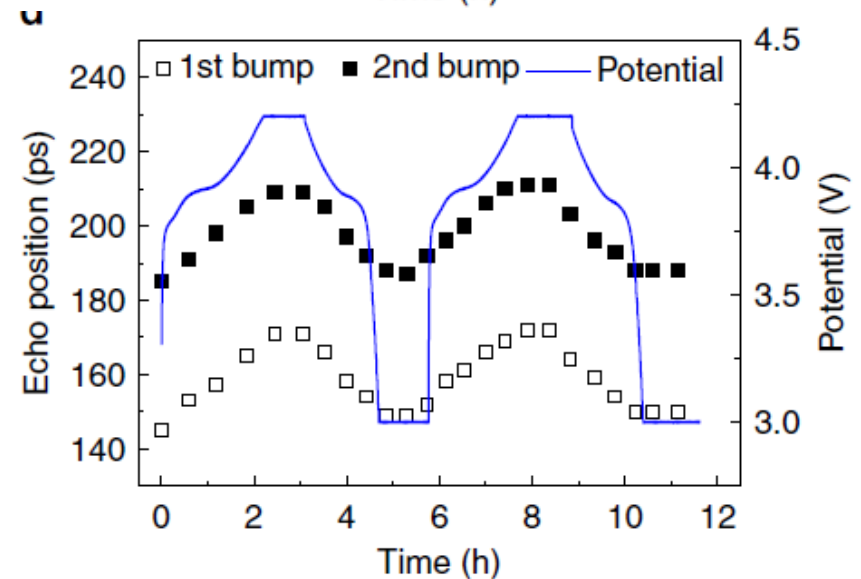
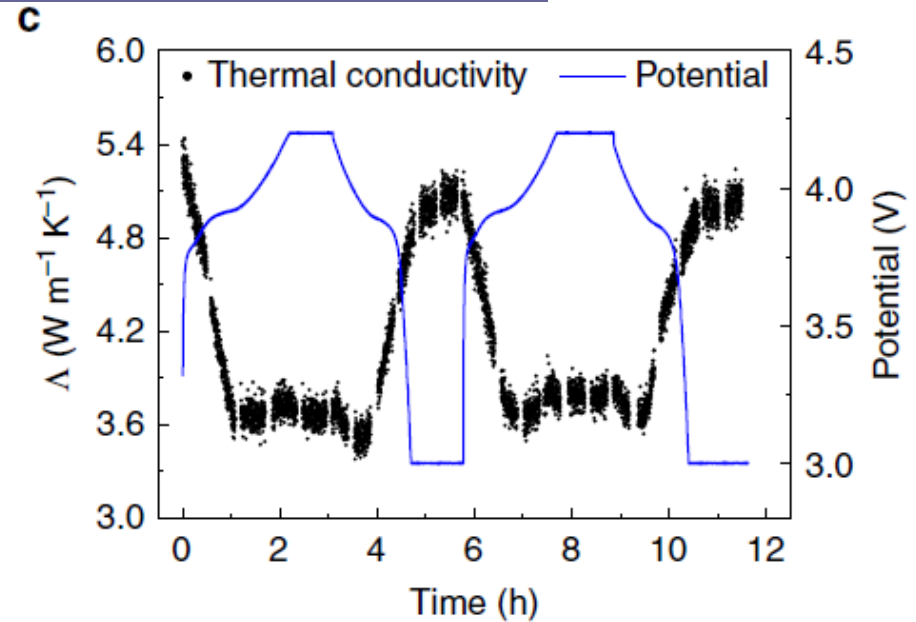
In-situ measurements of thermal conductivity and elastic constants

- Full delay time scans of $\text{Li}_{0.5}\text{CoO}_2$ and LiCoO_2



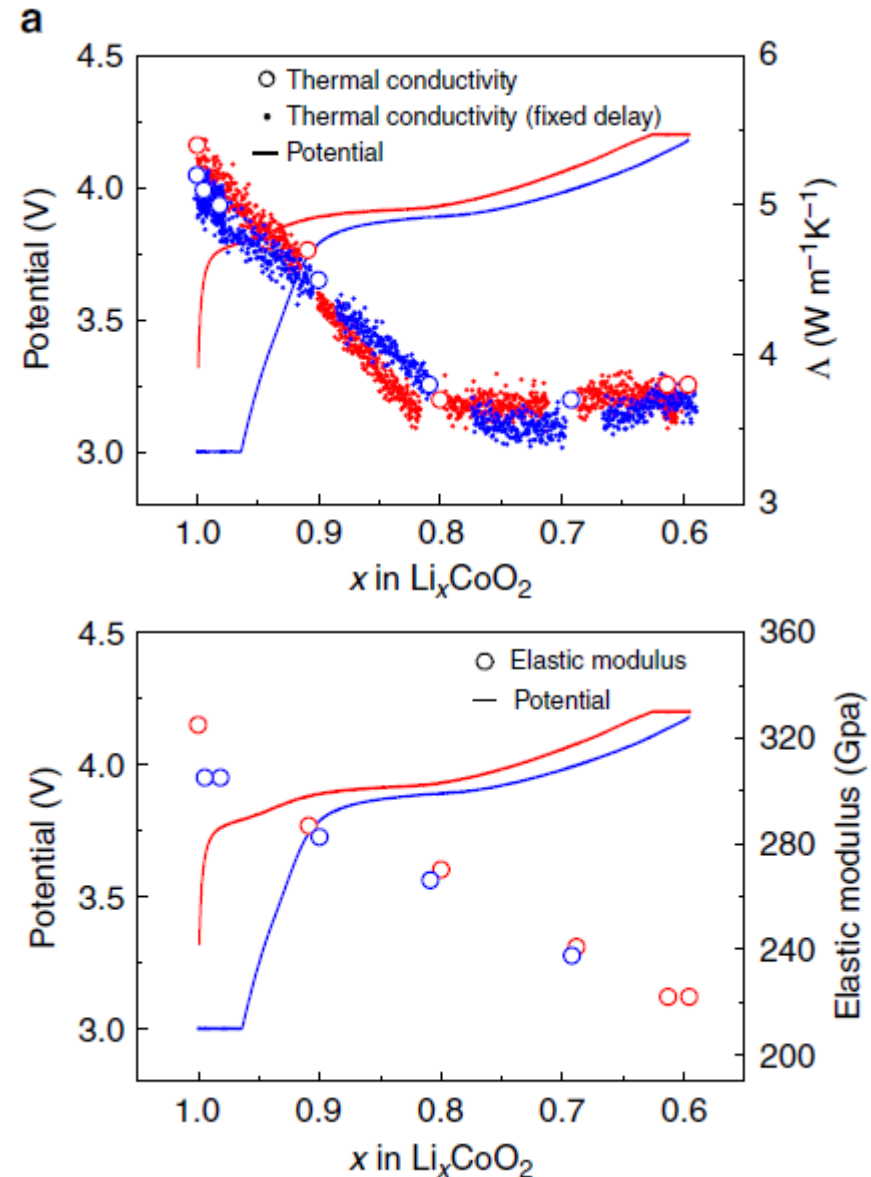
Continuous real-time measurements during electrochemical cycling

- With delay time set to a fixed value, ratio can be measured continuously and converted to thermal conductivity.
- Position of acoustic echo requires a scan over a limited range of delay times. Peak volume change is only 1.3% so changes in thickness are negligible.



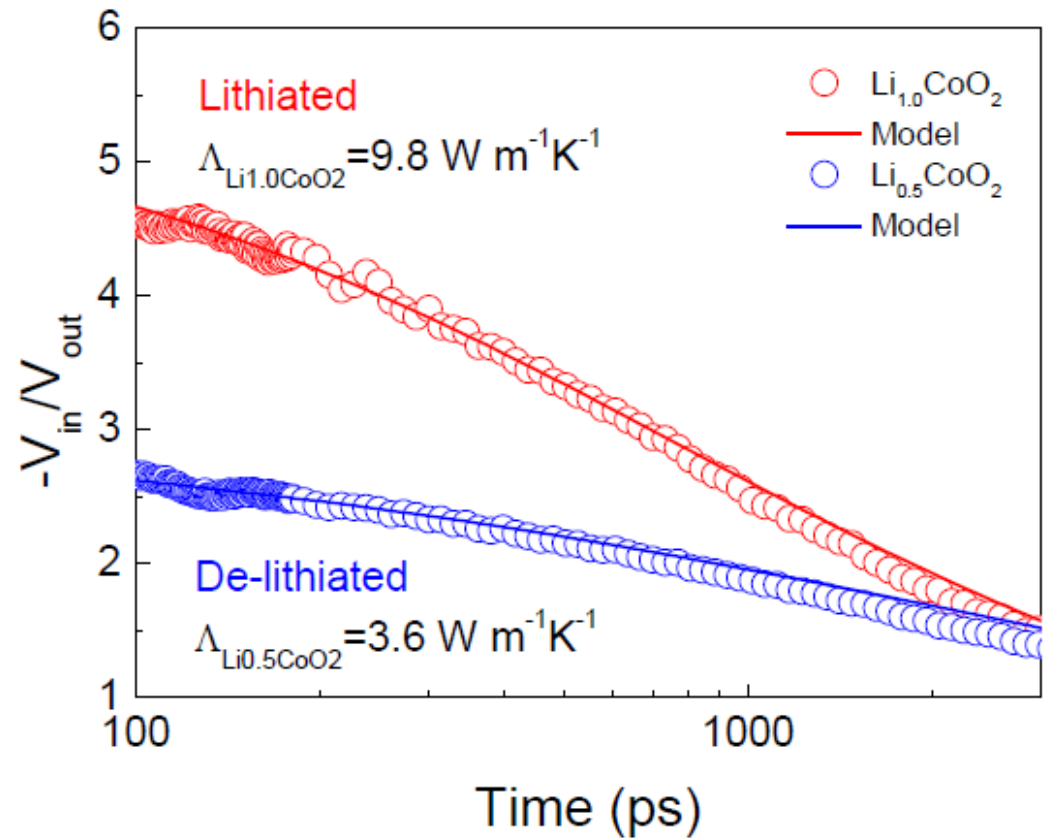
Continuous real-time measurements during electrochemical cycling

- Convert time-axis to composition. (We assume irreversible capacity loss occurs only during the lithiation cycle.)
- Thermal conductivity is not a linear function of x ; plateau for $0.5 < x < 0.8$
- Longitudinal elastic modulus is a linear function of x .



Ex-situ measurements of film annealed at 700°C shows higher conductivity in fully lithiated state.

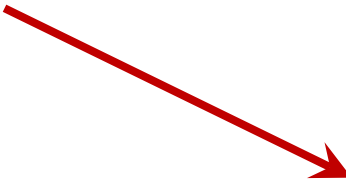
- Not yet sure of the mechanism.
 - Different texture?
 - Larger grain size?
 - Fewer point defects?



Do Li vacancies scatter phonons?

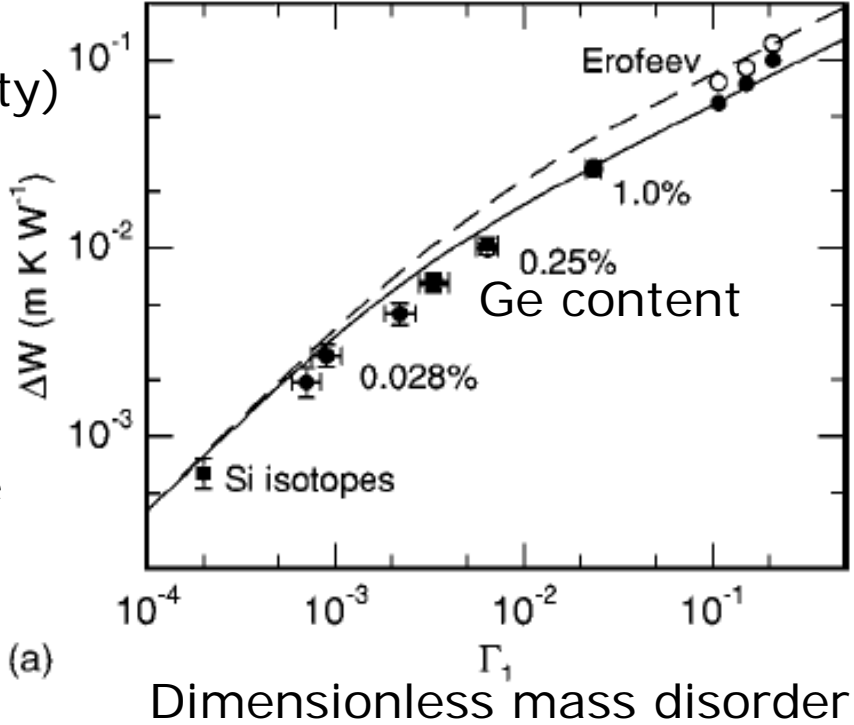
- Classic example of point defect scattering is mass disorder created by isoelectronic substitution, e.g., SiGe alloy

Change in thermal resistivity
(Reciprocal of thermal conductivity)



- Unlikely that random Li vacancies alone can explain the dependence of thermal conductivity on x .

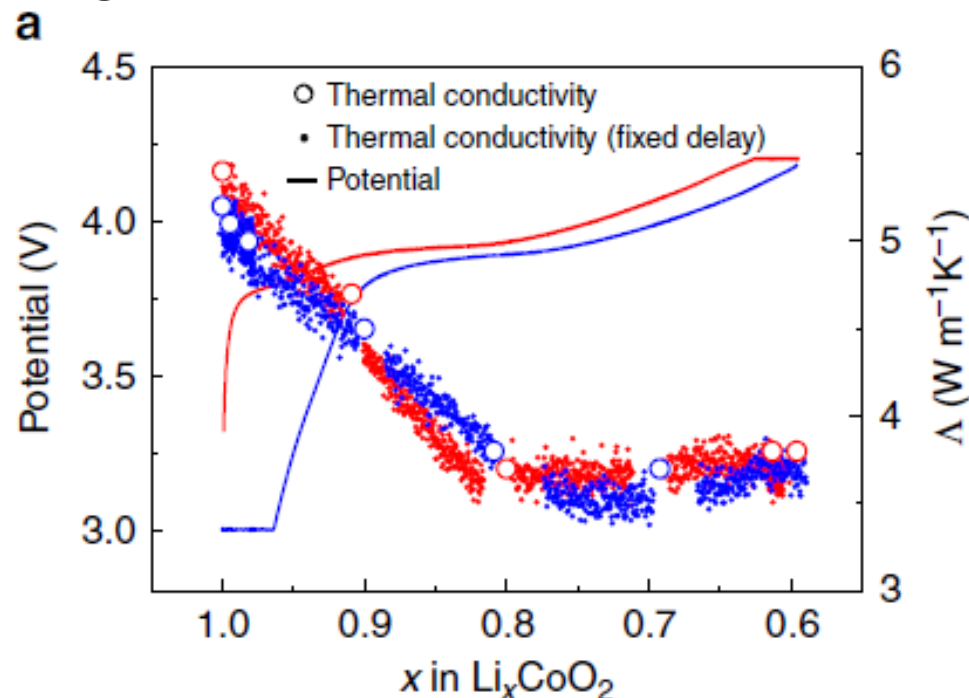
dilute SiGe alloys



(a)

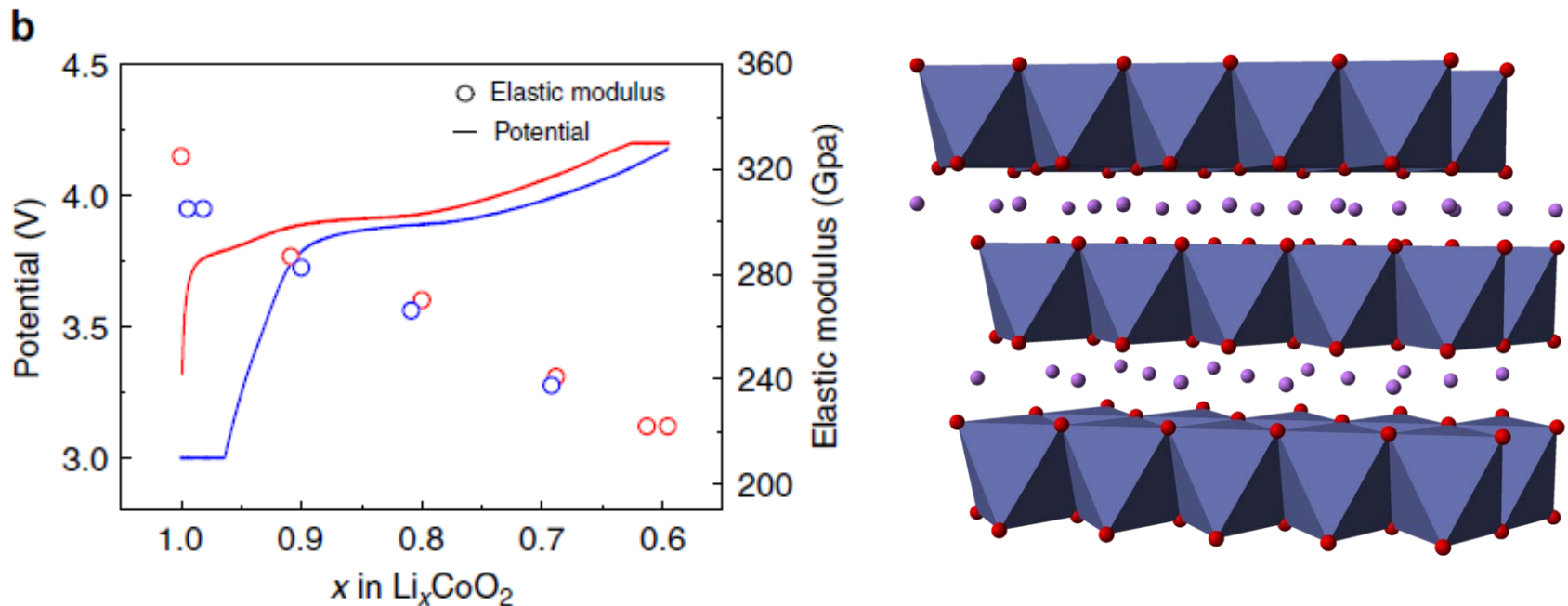
Mixture of Li rich and Li poor nanoscale phases?

- Evidence in the literature (Reimer *et al.*, JES (1992)) for a two-phase region $0.75 < x < 0.93$.
- This possibility makes the situation exceedingly complicated to predict the effect on thermal conductivity: disorder and characteristic size of each phase could vary with the average lithium content.



Li content has a strong influence on stiffness of bonds in the CoO_2 sheets

- Our samples are not textured so the change in longitudinal modulus is most due to C_{11} (stretch/compress along a - b plane)
- Higher Li content \rightarrow greater electron density in the CoO_2 sheets \rightarrow increased bond strengths (?)



Summary

- Time-domain thermoreflectance and picosecond acoustics enable real-time measurements of thermal conductivity and elastic constants of electrode materials.
- Contrast between low and high thermal conductivity states of Li_xCoO_2 up to a factor of 2.7.
- Working on getting full set of elastic constants: by experiment (surface-acoustic waves; orientation dependence) and theory (DFT by Prof. Elif Ertekin).
- Changes in longitudinal elastic modulus are linear in x ; i.e., virtual crystal or effective medium seems to apply.
- Changes in thermal conductivity are not linear in x and show a plateau for $0.5 < x < 0.8$.
 - Speculate that this is caused by changing mixture of phases.